

Microindentation characterization of polymers and polymer based nanocomposites

V. Lorenzo

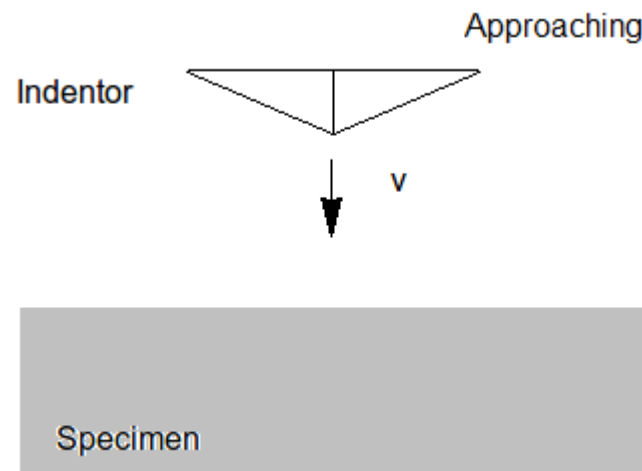
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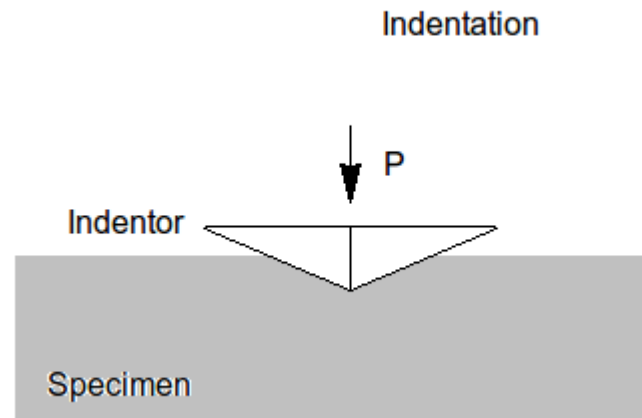
- Hardness and hardness measurement
 - Vickers hardness
 - Relationships between hardness and other mechanical properties of polymers
 - DSI
 - Microindentation and viscoelasticity
- Microhardness of heterogeneous polymer systems
 - Microhardness of semicrystalline polymers
 - Microhardness of blends
 - Microhardness and physical ageing
 - Microhardness of PMC's
 - Microhardness of PMnC's

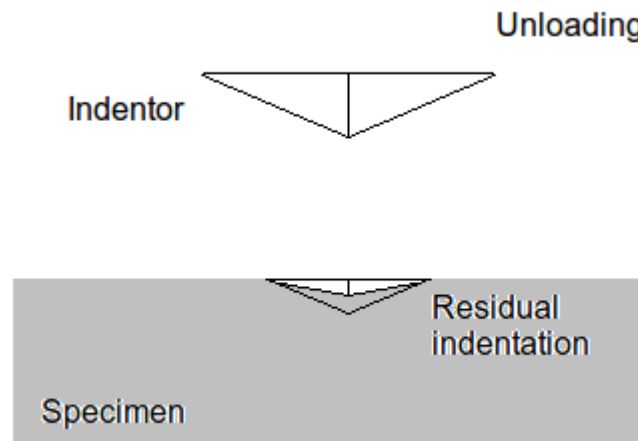
- **DEFINITION:** a measure of the resistance to permanent surface deformation or damage.
 - Local character of measurement
 - What is the meaning of surface damage?
- **METHODS OF TESTING:**
 - Scratching
 - Static indentation
 - Dynamic indentation
 - ...

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- Static penetration test
- Diamond square based pyramidal indenter (angle between the faces: 136°)
 - Diamond: indenter remains undeformed during the test
 - Pyramidal: geometric similarity of indentations \Rightarrow hardness is load independent
 - 136° : $HV \approx HB$ if $HB < 600$



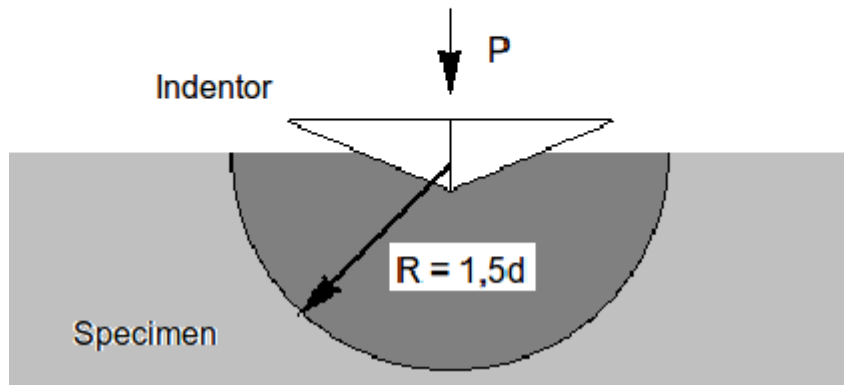




- Average pressure on the lateral surface of the residual indentation
- HYP.: the geometries of indenter and indentation are similar \Rightarrow
 - $h = d/7$ and $S_{\text{lat}} = d^2/(2 \cdot \sin 68^\circ)$

d: diagonal of the base of the residual imprint
h: indentation depth; S_{lat} : contact area
 - $HV = 2 \cdot \sin 68^\circ P/d^2$

HV: Vickers hardness; P: load
- MICROHARDNESS: hardness measured after applying small loads (grams) \Rightarrow diagonal of the residual indentation: μm

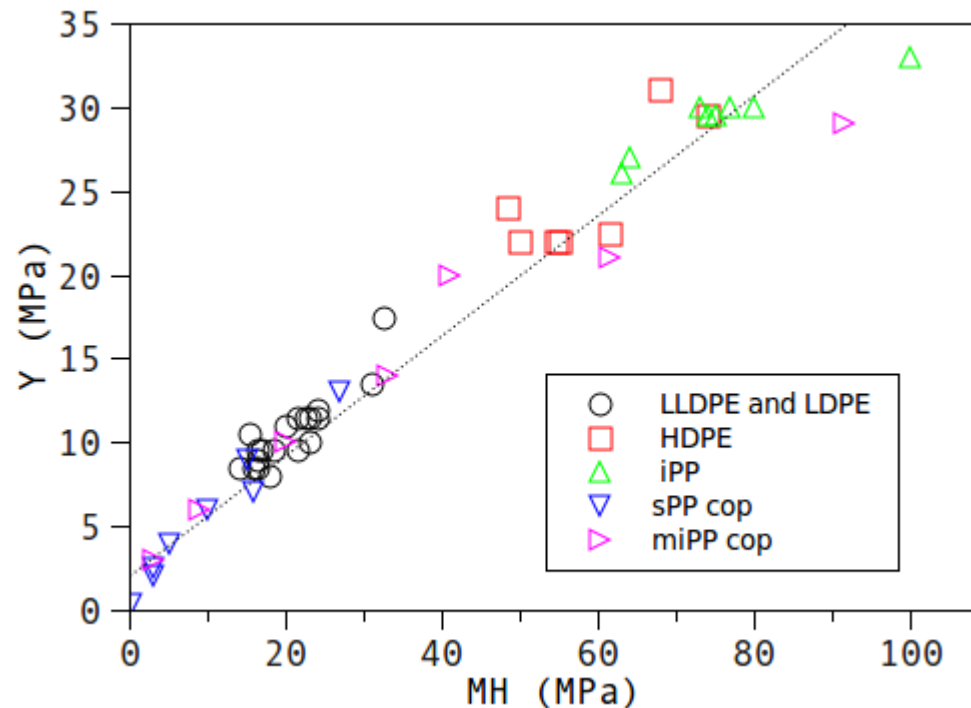


- Classical results and FEM calculations:
 - Stresses are confined to a hemispherical region with radius $\approx 1,5d \approx 10h$
- Some practical considerations:
 - Minimal distance between two indentations
 - Minimal distance between indentations and edges
 - Minimal thickness of films
 - Very small sample quantity (ng)

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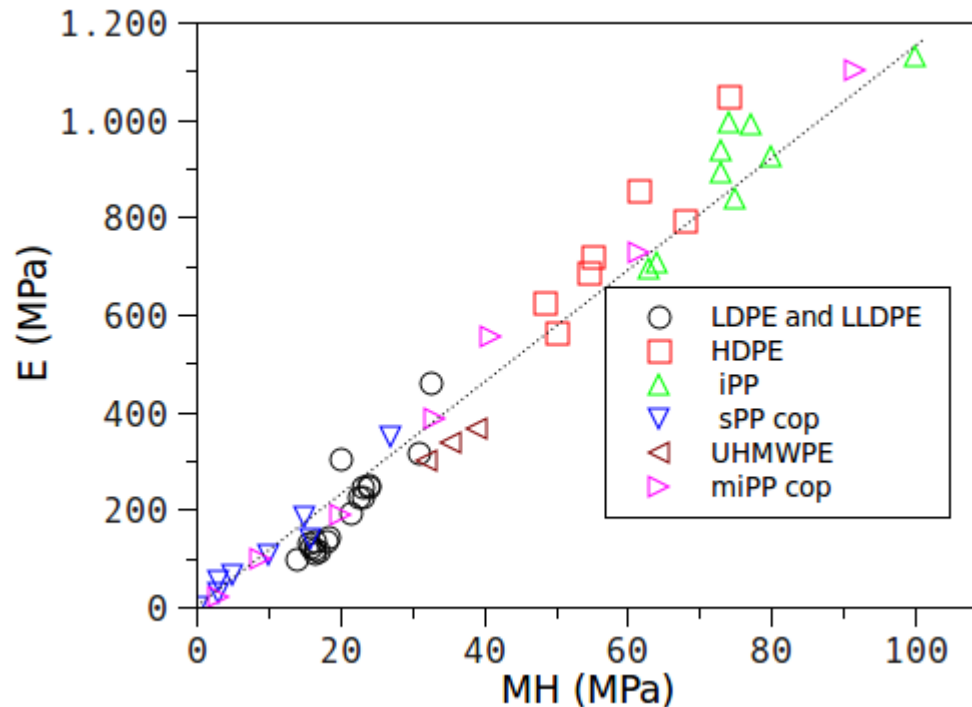
Relationships between MH and other mechanical properties



V. Lorenzo et al.: *Die Ang. Makromol. Chem.*, 172 (1989) 25-35

V. Lorenzo et al.: *J. Mater. Sci. Let.*, 8 (1989) 1455-1457

J. Arranz et al.: *Polymer*, 46 (2005) 12287-12297

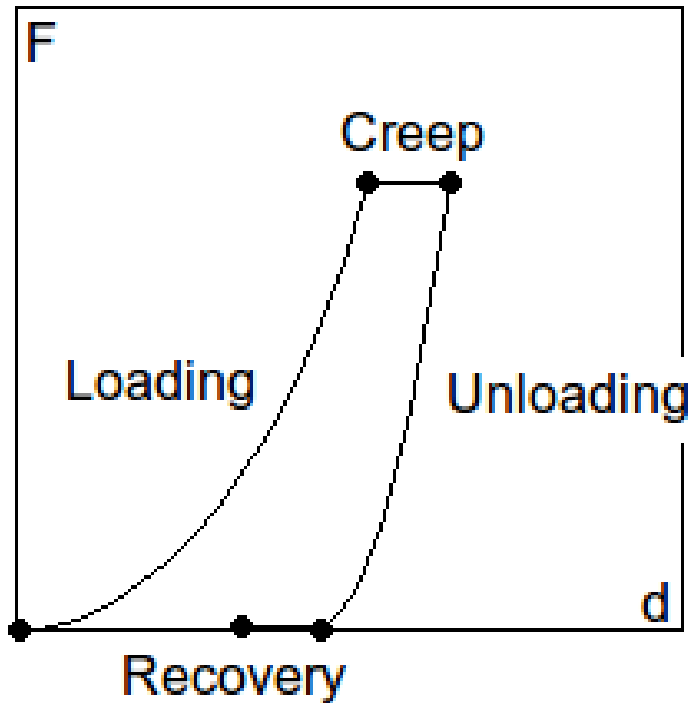


V. Lorenzo et al.: *Die Ang. Makromol. Chem.*, 172 (1989) 25-35

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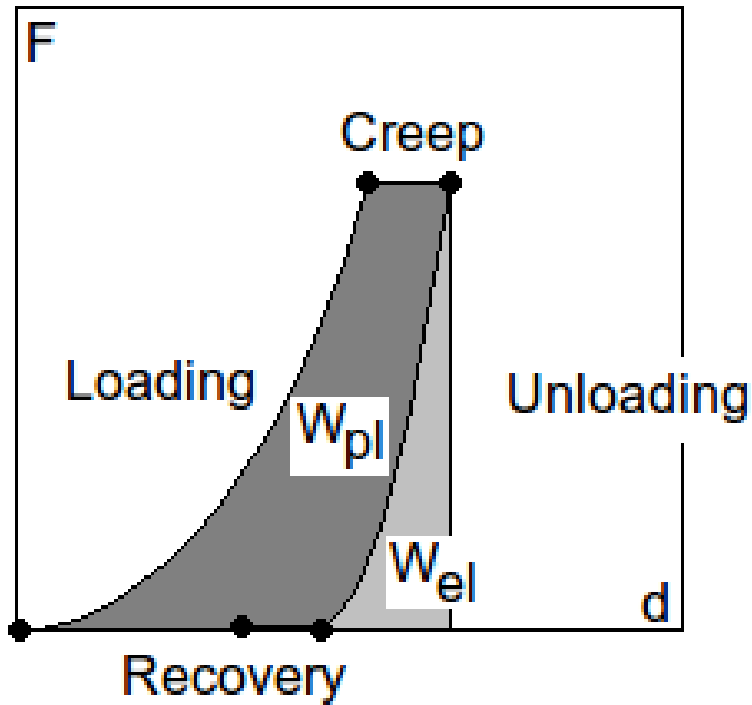
G. Zamfirova et al.: *J. Appl. Polym. Sci.*, 88 (2003) 1794-1798

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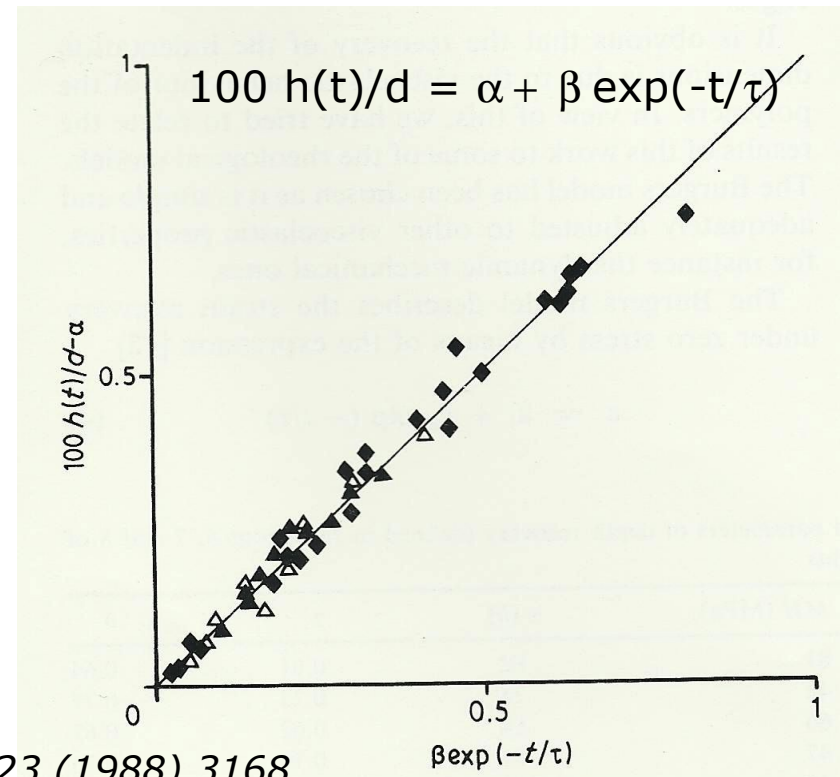
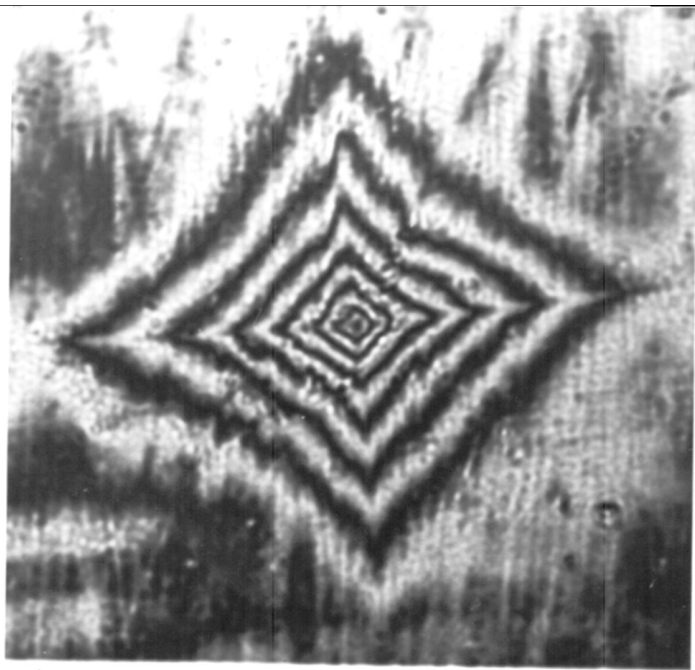
- 1980's: continuous measurements of load and indentation depth
- Very small loads (mN)
⇒ resolution: μN
- Very small indentation depths (tenths of μm) ⇒ resolution: nm
- Berkovich indenter

Fischer Cripps, A.: 'Nanoindentation', Springer (2004)



- Hardness under load
- Creep
- Elastic modulus
- Delayed elastic recovery
- Deformation energy
- Recoverable energy
- ...
- And, of course, hardness

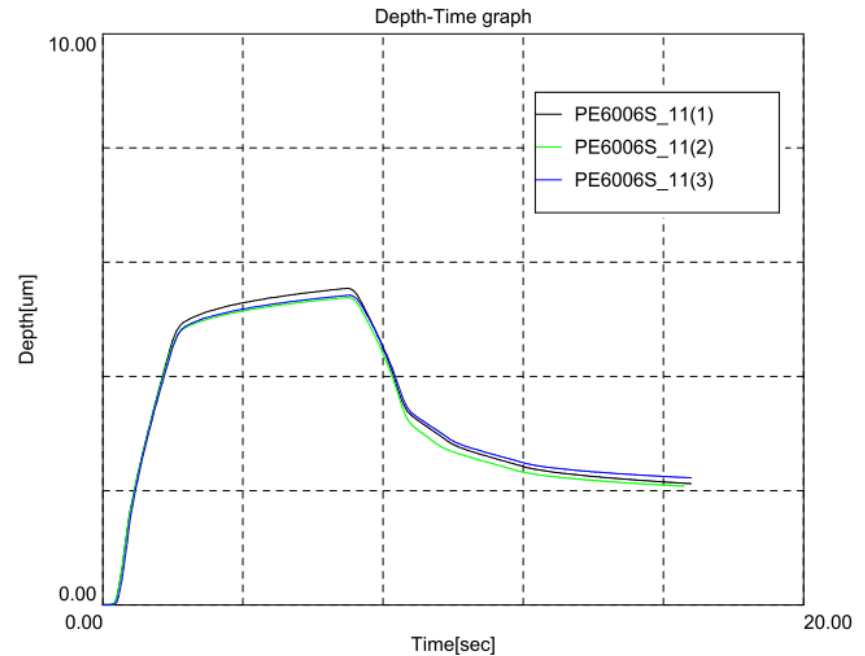
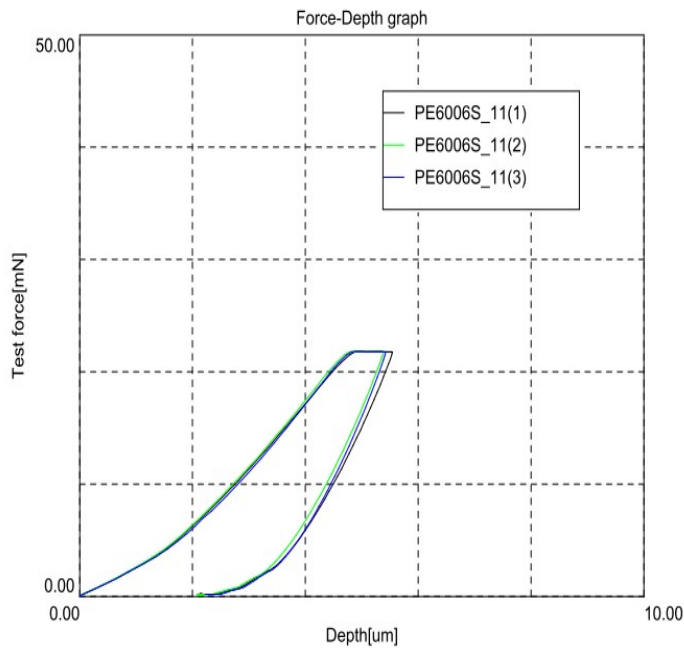
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V.Lorenzo et al.: J. Mater. Sci., 23 (1988) 3168



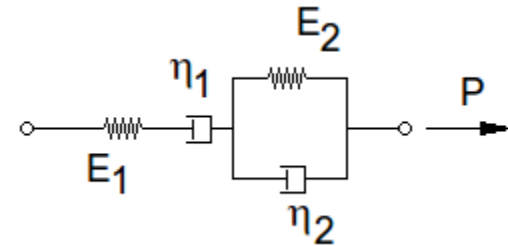
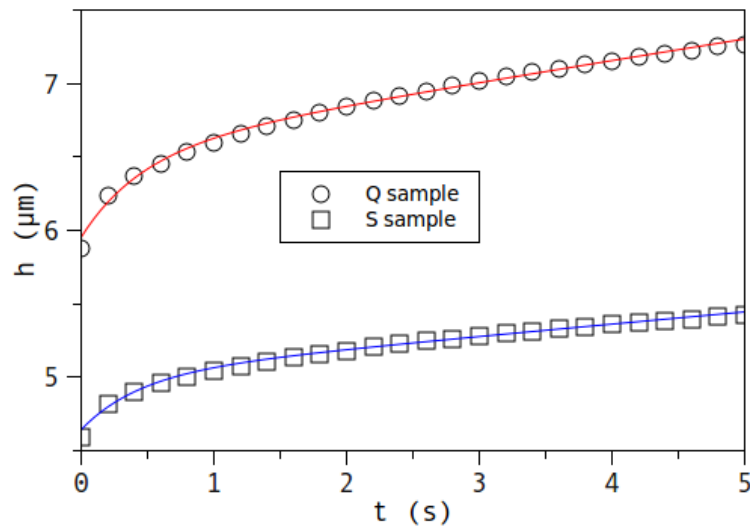
Delayed elastic recovery study by means of DSI measurements



V.Lorenzo et al.: communication to EPF2011, Granada, 26th June-1st July



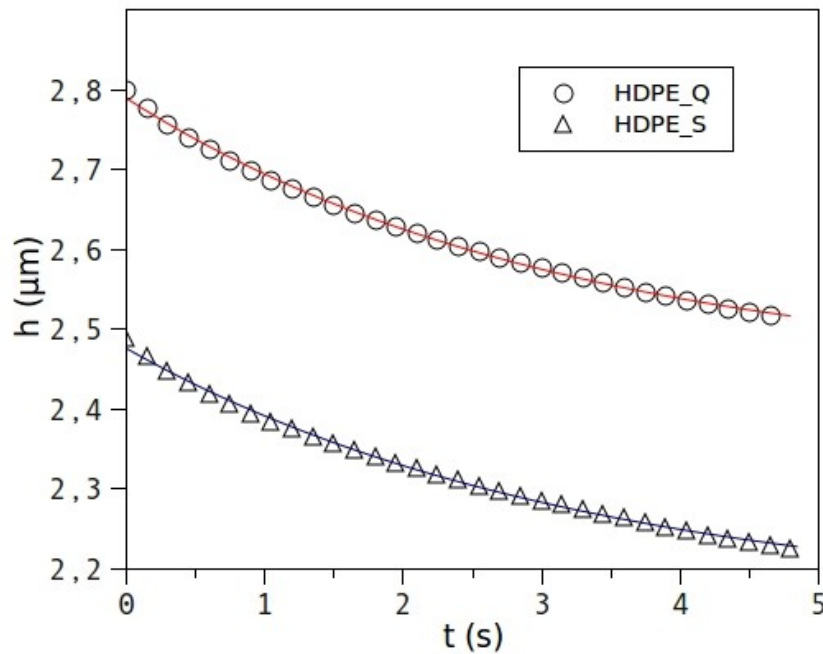
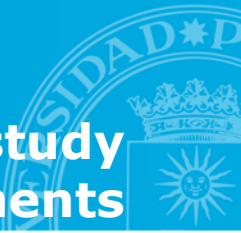
- Model:



$$h^2(t) = \frac{\pi}{2} P_0 \cotg \alpha \left[\frac{1}{E_1} + \frac{1}{E_2} \left(1 - \exp\left(-\frac{E_2 t}{\eta_2} \right) \right) + \frac{t}{\eta_1} \right]$$

Sample	E (MPa)	τ (s)	Cit (%)
Q	410	0,48	23
S	675	0,46	18

V.Lorenzo et al.: communication to EPF2011, Granada, 26th June-1st July



Model:

$$h(t) = h(0) + A(1 - \exp(-t/\tau))$$

- Retardation time, τ , structure independent
- Not a quantitative agreement
 - Interferometry experiments
 - Indenter geometry
 - Sampling frequency
 - Creep results:
 - Non linearity

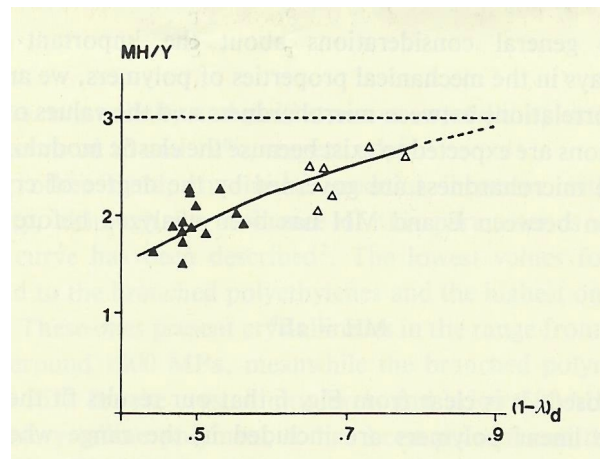
V.Lorenzo et al.: communication to EPF2011, Granada, 26th June-1st July

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- Microindentation “averages” the properties of some μm^3 of the material around the indenter.
- Microindentation and heterogeneity of the specimen:
 - Characteristic length of heterogeneities $> d \Rightarrow MH = f(x, y)$
 - Information about distribution of phases
 - Characterization of phases
 - Characteristic length of heterogeneities $< d \Rightarrow MH$ is not a function of the position
 - Bulk properties of the material

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- Polyethylenes that has been quickly cooled form the melt:
 - Length of crystallites $< d \Rightarrow$ MH is not a function of the position \Rightarrow MH is an increasing function of crystallinity level
 - Information about deformation mechanism



V. Lorenzo et al.: *Die Ang. Makromol. Chem.*, 172 (1989) 25-35



- It is not possible to obtain a 100% crystalline or amorphous PE sample.
- Mechanical properties of phases can be obtained by extrapolating MH measurements

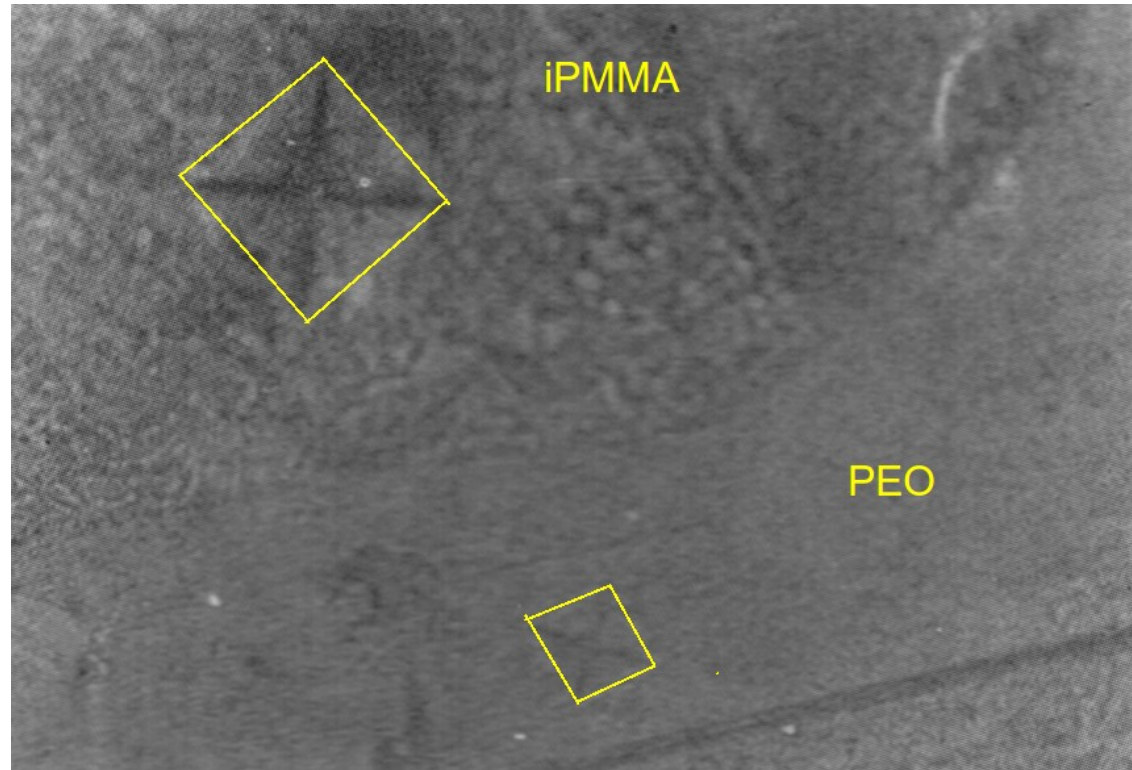
*Baltá-Calleja, F.J.; Fakirov, S., 'Microhardness of Crystalline Polymers' ;
Cambridge University Press: Cambridge, 2000*

- Isothermally crystallized iPP displays spherulites of the α and β polymorphs \Rightarrow properties of the α and β spherulites

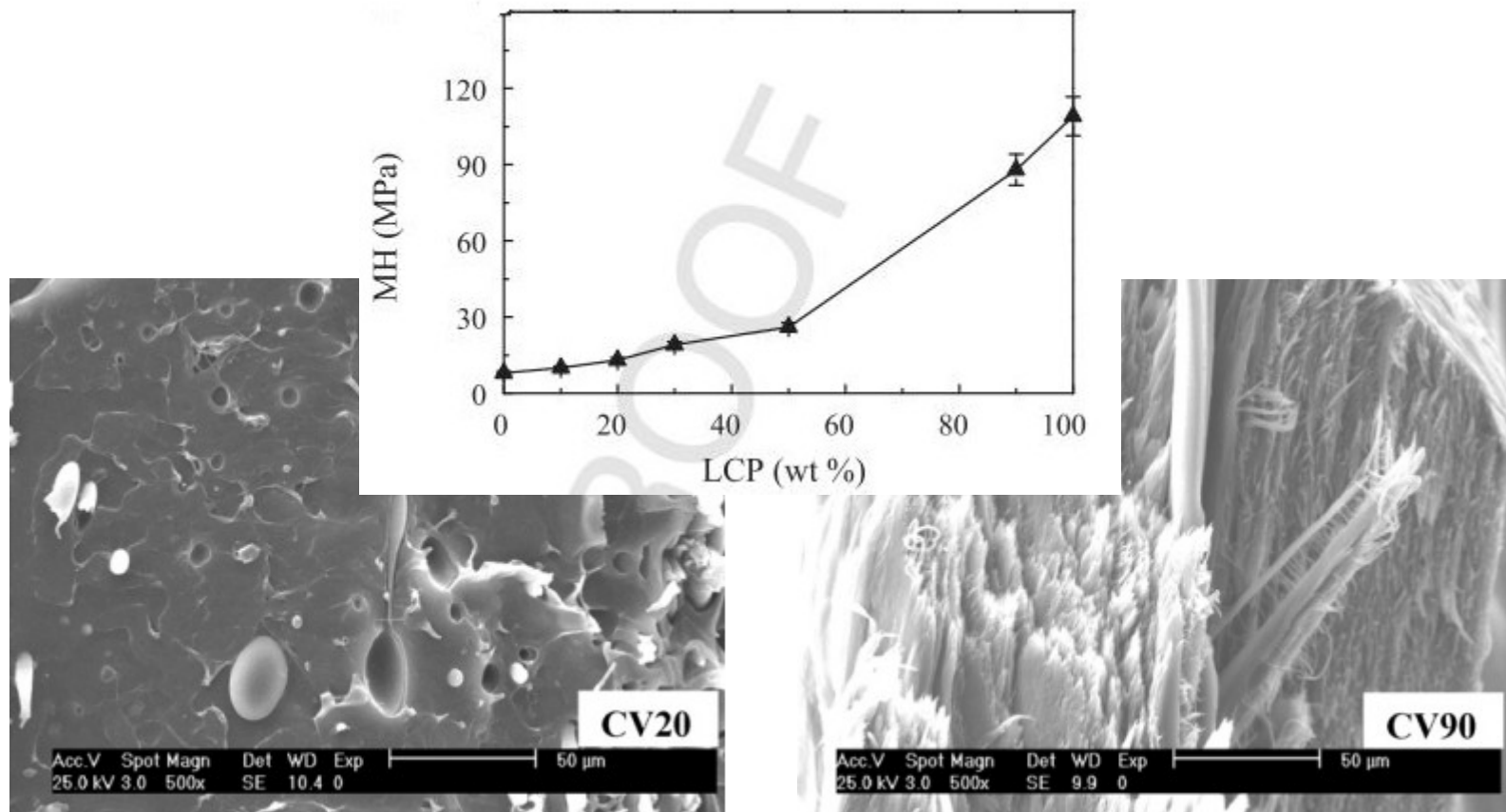
*Seidler, S.: JOURNAL OF MACROMOLECULAR SCIENCE, Part B—Physics, B41
(2002) 851–861*

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- Blends of miscible A and B polymers: homogeneous at d scale \Rightarrow $MH = f(\%A)$
- Blends of immiscible A and B polymers: separated domains of A and B
 - If $\%A \ll \%B \Rightarrow$ characteristic length of A domains $< d \Rightarrow$ MH is a continuous function of $\%A$
 - Continuity of $MH(\%A) \Rightarrow$ miscibility
 - If $\%A$ is comparable con $\%B$:
 - Characteristic length of A domains $< d$
 - Characteristic length of A domains $> d$
 - MH is a function of the position
 - Characterization of individual phases



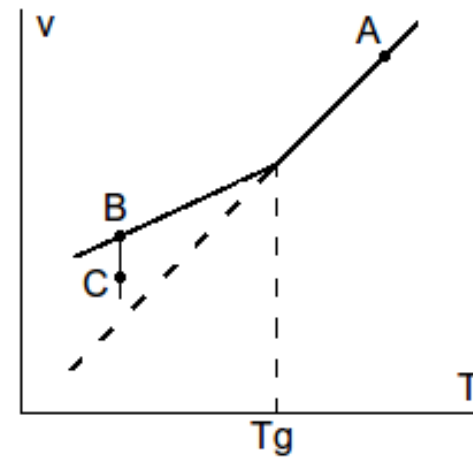
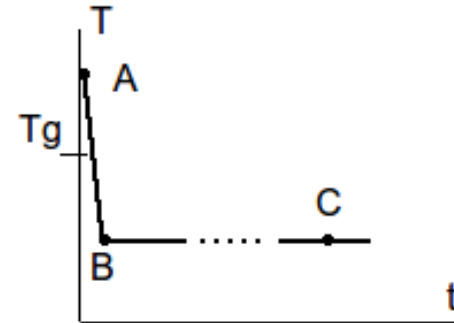
V. Lorenzo et al.: J. Mater. Sci. Lett., 9 (1990) 1011-1013



J. Arranz et al.: doi:10.1016/j.memsci.2011.04.023

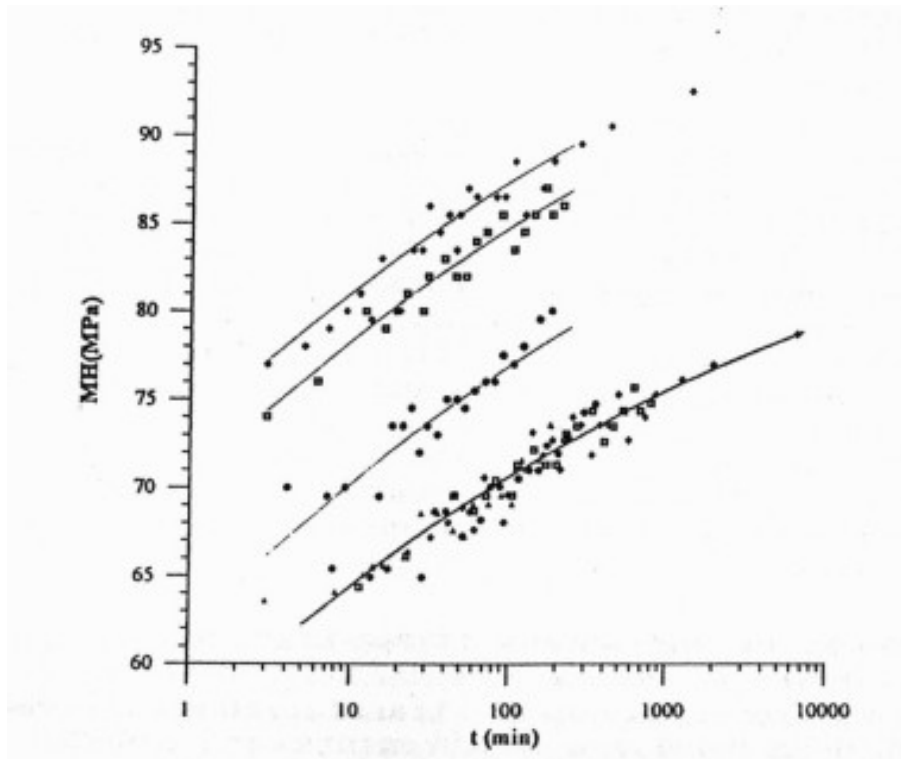
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- A – T_g: liquid
 - Cooperative movement of chains
- T_g – B: glass
 - Movements of local groups
- B – C: physical ageing
 - Densification:
 - Local free volume fluctuations
 - Correlation length < 10⁻¹ μm

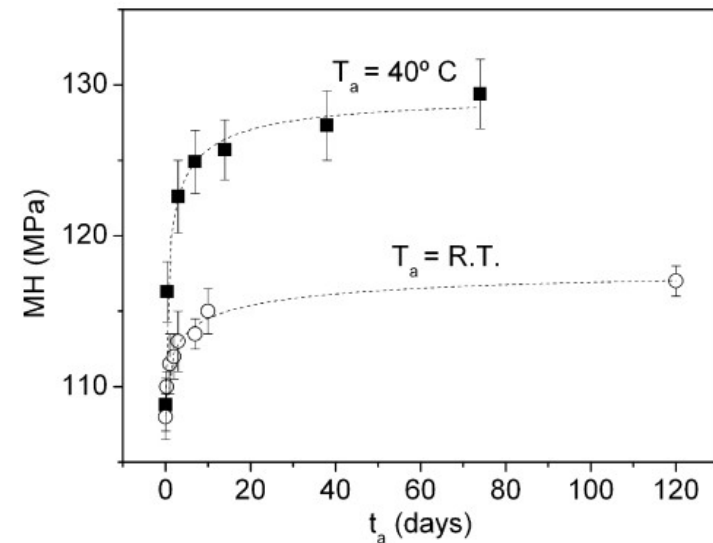




Physical ageing of LCP's and SMP's as revealed by MH tests



A. Ormazábal et al. In "Nanostructured and Non-Crystalline Materials", World Scientific, Singapore (1995) 202-206



V. Lorenzo et al.: *Materials and Design* 30 (2009) 2431–2434

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- Composites are multiphasic materials: fillers dimensions \sim some tens of μm \Rightarrow characteristic dimensions of heterogeneities $> d \Rightarrow$ MH is position function \Rightarrow MH is not an adequate tool for characterizing composite materials

But it can be used for:

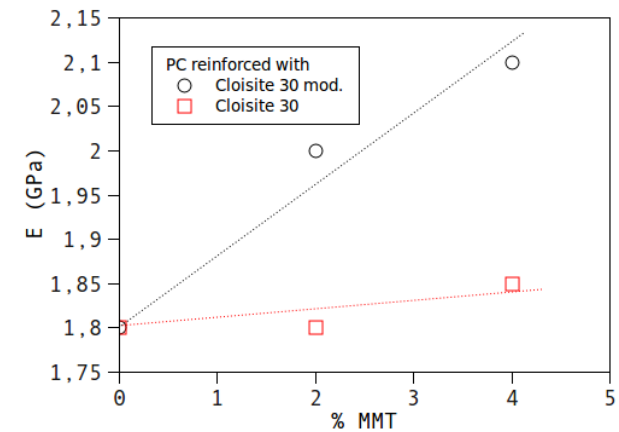
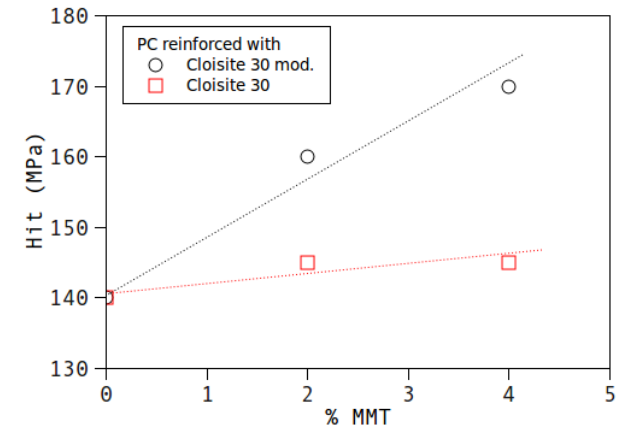
- Characterizing matrix and fillers.
- Characterizing interphases:
 - Transcrystalline structures in GF reinforced PP composites

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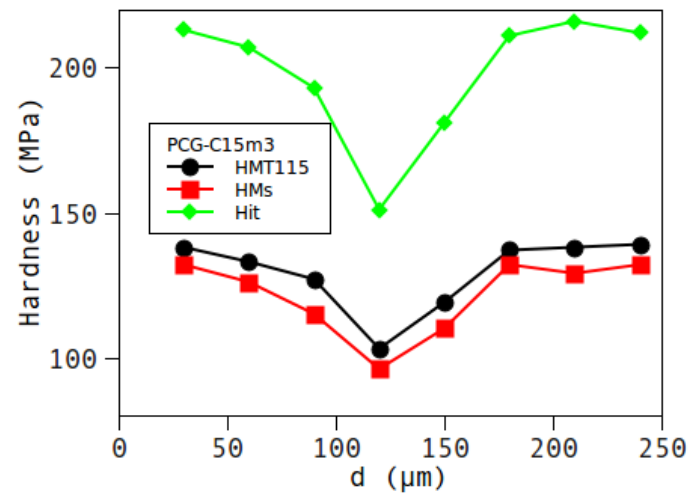
- Agglomerated fillers:
 - characteristic length of heterogeneities $> d \Rightarrow$ MH is a function of position
- If the fillers are well dispersed:
 - characteristic length of heterogeneities $< d \Rightarrow$ MH = f(% filler)
 \Rightarrow information about the reinforcement effect of the filler.



PC-clay nanocomposites obtained by dissolution



- Characterization of coatings
- Multi-layer extrusion
- Skin-core structures in injection molded polymers
- Composition gradients
- ...



- Microindentation is an adequate tool for exploring structure of polymeric materials
- The volume of material that is deformed in hardness test is around d^3
- The information that can be obtained from a hardness test depends on the characteristic length of the heterogeneities of the sample, l :
 - If $l < d$, bulk properties of the material
 - If $l > d$, local character information

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 - Prof. E. Pérez
 - Prof. M.L. Cerrada
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- ETSII (UPM)
 - Prof. J. Martínez
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 - Prof. G. Pinto
 - Prof. V. Lorenzo

and all the postdoc, students and technicians that have collaborated with us.



Thank you very much for your attention

Merci pour votre attention

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